

Magnesium borohydride: A new hydrogen storage material

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Abstract

Magnesium borohydride ($\text{Mg}(\text{BH}_4)_2$) is a promising material for hydrogen storage because of its high gravimetric storage density (15.0 mass%). We intended to synthesize $\text{Mg}(\text{BH}_4)_2$ by decomposition reaction of LiBH_4 with MgCl_2 by heat treatment without using a solvent, where the product consists of LiCl and a compound of magnesium, boron and hydrogen. Hydrogen desorption temperature of the product is approximately 100 K lower than that of LiBH_4 and the decomposition consists of a two-step reaction. The products of the 1st and 2nd decomposition reactions are MgH_2 and Mg , respectively. This result indicates the following two-step reaction (1st reaction: $\text{Mg}(\text{BH}_4)_2 \rightarrow \text{MgH}_2 + 2\text{B} + 3\text{H}_2$, 2nd reaction: $\text{MgH}_2 \rightarrow \text{Mg} + \text{H}_2$). The first decomposition peak is dominant and is around 563 K. The 2nd decomposition occurs at the temperature greater than 590 K.

Keywords: Hydrogen absorbing materials; Thermodynamic properties; Thermal analysis

1. Introduction

High-density hydrogen storage is one of the most important issues to introduce fuel cell vehicles. Complex hydrides, consisting of light elements, are very promising materials for hydrogen storage because of their high gravimetric and volumetric hydrogen density [1]. Since Bogdanovic and Schwickardi reported the reversibility of the catalyzed hydrogen sorption reaction of NaAlH_4 [2], many efforts have been made to investigate complex hydrides as hydrogen storage materials [3–5]. Besides the basic physical properties, the challenges are to tailor the stability and to investigate the reversibility of the hydrogen sorption reaction of complex hydrides.

The hydrogen in the complex hydrides is often located in the corners of a tetrahedron with B or Al in the center. The negative charge of the anion, $[\text{BH}_4]^-$ and $[\text{AlH}_4]^-$, is compensated by a cation, e.g. Li or Na. Therefore, the bonding character and the properties of the complex hydrides are largely determined by the difference in

electronegativity between the cation and the boron or aluminum. For example, the decomposition temperatures of complex hydrides consist of alkaline metal and $[\text{BH}_4]^-$ or $[\text{AlH}_4]^-$ anion have a good correlation with electronegativity of cation as is shown in Fig. 1. Based on this correlation, experimental studies have been performed to lower the stability of complex hydrides [6–8]. Also, theoretical calculations indicate that this tendency can be applied not only to alkaline metal borohydride, but also to other borohydrides with alkaline-earth metals or some of transition metals [9,10].

Magnesium borohydride ($\text{Mg}(\text{BH}_4)_2$) is one of the promising materials for hydrogen storage because of its high gravimetric storage density (15.0 mass%). The Pauling electronegativity of magnesium is 1.31, which is greater than that of lithium (0.98). This implies that $\text{Mg}(\text{BH}_4)_2$ is less stable than LiBH_4 . Until now, physical properties of $\text{Mg}(\text{BH}_4)_2$ as a hydrogen storage material are not known because it is difficult to synthesize the material. The process to synthesize $\text{Mg}(\text{BH}_4)_2$ has been mainly examined by two different approaches. One is the reaction of diborane (B_2H_6) with magnesium or its compounds [11]. The other is the double decomposition of magnesium halides with

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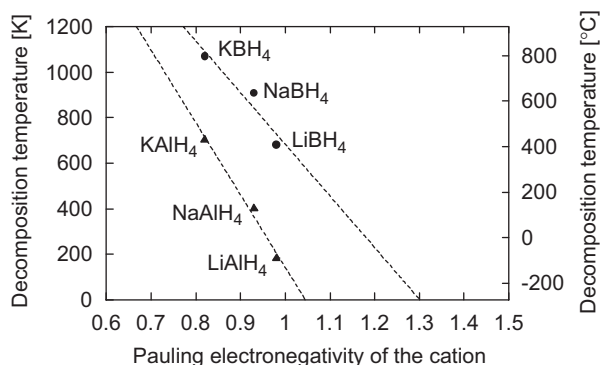


Fig. 1. Decomposition temperature as a function of Pauling electronegativity of the cation.

alkaline metal borohydrides in an organic solvent, e.g. diethyl ether or tetrahydrofuran [11,12]. The second approach has the advantage that it does not require diborane, which is a toxic compound. However, the product which is synthesized by double decomposition in a solvent is a solvate consisting of $\text{Mg}(\text{BH}_4)_2$ and solvent (e.g. $\text{Mg}(\text{BH}_4)_2 \cdot 3\text{THF}$) and the extraction of $\text{Mg}(\text{BH}_4)_2$ from the solvate is difficult because of its affinity with the solvent [13].

In this study, we report about a new method intended to synthesize $\text{Mg}(\text{BH}_4)_2$ by dry process without using a solvent. Lithium borohydride and magnesium chloride are used as starting materials. The synthesis products as well as the hydrogen desorption products were investigated by X-ray diffraction and hydrogen desorption measurement of the product synthesized by the new method have been performed.

2. Experimental

The samples were purchased from Aldrich Co. Ltd.: the purities are >95% for LiBH_4 , >99.9% for MgCl_2 , respectively. The samples were always handled in an argon glove box to avoid any possible reaction with moisture or air. The reaction between LiBH_4 and MgCl_2 was investigated by differential scanning calorimetry (DSC) (Mettler Toledo Inc. HP DSC827e). Five milligrams of the sample (for 2:1 mole ratio) was mixed in an argon glove box and filled in a DSC sample cell made of aluminum. The sample cell was sealed up in argon atmosphere. Therefore, it has never been exposed to any gases except pure argon during the measurement. DSC measurement of LiBH_4 and MgCl_2 mixture was carried out in the temperature range from 313 to 513 K at the heating (or cooling) rate of 5 K/min in the sealed sample cell for three heating and cooling cycles. Measurements with pure LiBH_4 and MgCl_2 were also performed at the same heating condition.

Heat treatments of LiBH_4 and MgCl_2 mixture were carried out in a stainless steel cylinder <10 MPa of hydrogen at 453, 523 or 593 K. In each heat treatment, approximately 600 mg of LiBH_4 and MgCl_2 mixture (for 2:1 mole ratio) was pressed in order to make a pellet and

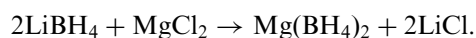
filled into a cylinder in an argon glove box. After evacuating by rotary vacuum pump for 1 h at room temperature, 10 MPa of hydrogen was introduced into the cylinder and heated up. It was kept at the final temperature mentioned above for 3 h and then slowly cooled to room temperature.

Crystal structures of the samples were investigated at room temperature by powder X-ray diffraction measurement (SIEMENS, D-500, Cu K α). To avoid exposure to air, each sample was filled into a sample holder in an argon glove box and covered with plastic wrap film during X-ray diffraction measurement.

Temperature-programmed desorption (TPD) measurement was carried out in vacuum continuously after heat treatment at 593 K in the same stainless steel cylinder. The desorbed gas volume was measured by a mass flow controller (Brooks instruments, 5850 E, max. flow 5 standard cm^3/min), with a maximum full-scale error of 1%.

3. Results and discussion

The DSC profiles of LiBH_4 and MgCl_2 mixture are shown in Fig. 2. Profiles of pure LiBH_4 and pure MgCl_2 were also investigated as references. For LiBH_4 , there is an endothermic peak (at $T = 386 \text{ K}$, $\Delta Q = -206.0 \text{ J/g}$) during heating from 313 to 513 K and an exothermic peak during cooling both of them are corresponding to the phase transition of LiBH_4 , whereas no peak was observed for MgCl_2 in this temperature range. For the LiBH_4 and MgCl_2 mixture sample, there is an endothermic peak (at $T = 385 \text{ K}$, $\Delta Q = -56.3 \text{ J/g}$) during the 1st heating procedure, corresponding to the phase transition of LiBH_4 . However, this peak disappears after the cooling of 1st cycle. This result indicates that there is no pure LiBH_4 remains after the 1st heating. In addition, for the LiBH_4 and MgCl_2 mixture, other exothermic and endothermic peaks are observed after the 1st cooling at $T = 440 \text{ K}$ ($\Delta Q = 23.6 \text{ J/g}$). These peaks are not observed for pure LiBH_4 or pure MgCl_2 . This result implies that some reaction took place during the 1st heating. The endothermic peak at 440 K observed after the 1st cooling is in agreement with the DSC measurement result which Stasinevich and Egorenko has reported for $\text{Mg}(\text{BH}_4)_2$ [14]. Therefore, the following reaction during the 1st heating procedure is assumed:



XRD measurement results after heat treatment also indicate that LiBH_4 react with MgCl_2 during heat treatment. Fig. 3 shows the XRD measurement results after heat treatment at various temperatures. Simply mixing LiBH_4 and MgCl_2 in an inert atmosphere (Ar) does not lead to any reaction. However, after heat treatment, the peaks corresponding to LiBH_4 and MgCl_2 disappear. New peaks were observed which correspond to LiCl . Although there is still a small amount of LiBH_4 and MgCl_2 after heat treatment at 453 K, neither LiBH_4 nor MgCl_2 was observed after heat treatment at the

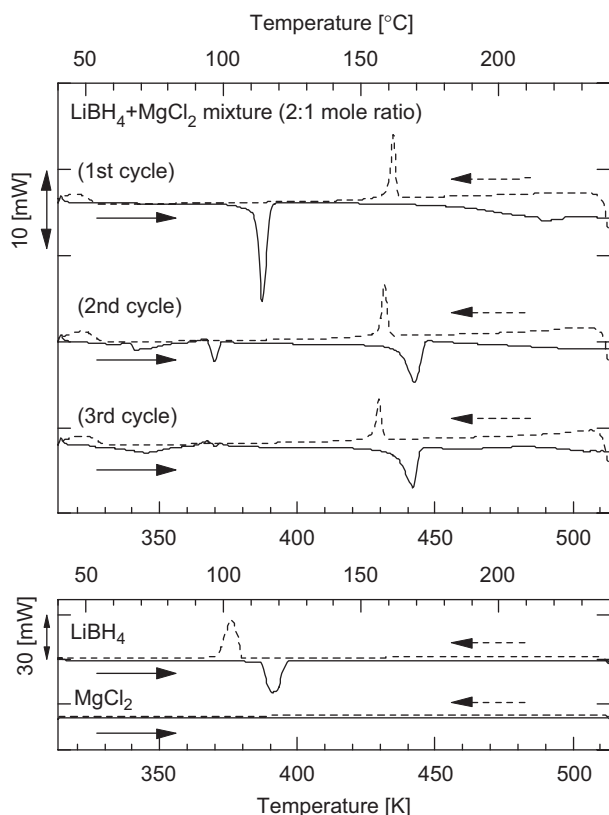


Fig. 2. Differential scanning calorimetry profiles of LiBH_4 and MgCl_2 mixture with a sealed sample cell. The heating rate is 5 K/min. Measurements with pure LiBH_4 and MgCl_2 were also performed at the same heating condition. Solid lines and dotted lines show heating and cooling, respectively. Five milligrams of the sample was used for each measurement.

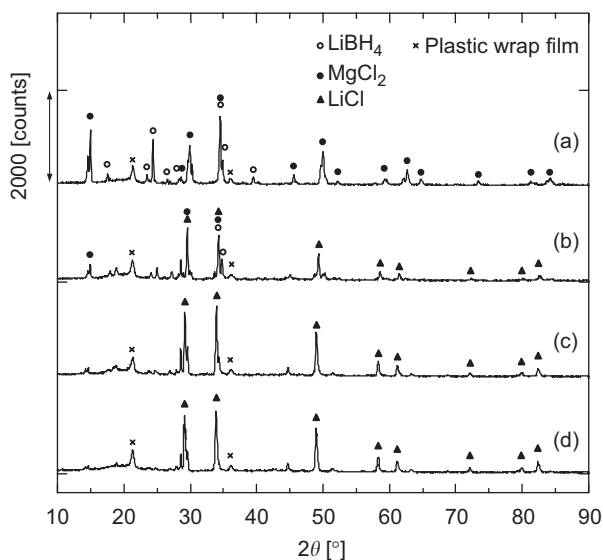


Fig. 3. Powder X-ray diffraction profiles of LiBH_4 and MgCl_2 mixture: (a) before heat treatment; (b) after heat treatment at 453 K; (c) after heat treatment at 523 K; and (d) after heat treatment at 593 K, respectively. Each heat treatment was carried out <10 MPa of hydrogen.

temperature > 523 K. The reason for the absence of a diffraction pattern for $\text{Mg}(\text{BH}_4)_2$ may be that the compound synthesized by this method does not crystallize sufficiently. Orimo et al. [15] have reported that long-range order of LiBH_4 disappears when LiBH_4 melts and solidifies again. The same phenomena may happen in the sample after heat treatment.

TPD measurement of the LiBH_4 and MgCl_2 mixture was carried out after heat treatment at 593 K for 3 h <10 MPa of hydrogen. After heat treatment, the sample was cooled down to room temperature <10 MPa of hydrogen. Subsequently, hydrogen gas was extracted from the cylinder to vacuum at room temperature, where no gas evolution was observed from the sample during this procedure. After evacuation, the sample was heated with a heating rate of 0.2 K/min. The thermograms of the TPD measurement are shown in Fig. 4. The hydrogen desorption temperature of the sample after the heat treatment is approximately 100 K lower as compared to that of the pure LiBH_4 sample. The sample after heat treatment has two desorption peaks. It starts to decompose at 500 K and the 1st desorption peak appears around 563 K. This peak is dominant and more than half of hydrogen is evolved. The second desorption peak starts at 600 K and it is very sharp as compared to the 1st desorption peak. The shape of two independent desorption peaks implies that decomposition reaction consists of two steps.

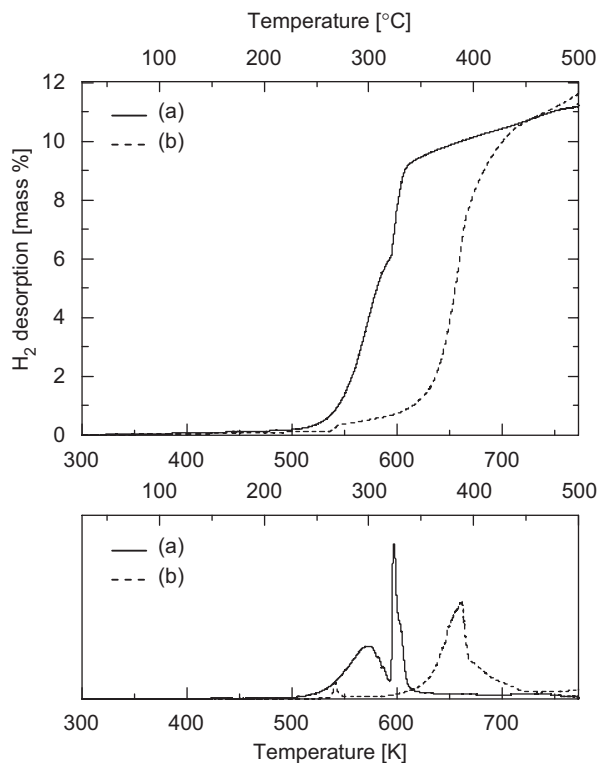


Fig. 4. Temperature-programmed desorption spectra of: (a) LiBH_4 and MgCl_2 mixture after heat treatment at 593 K <10 MPa of hydrogen; and (b) pure LiBH_4 . The samples were heated after evacuation at room temperature with a heating rate of 0.2 K/min.

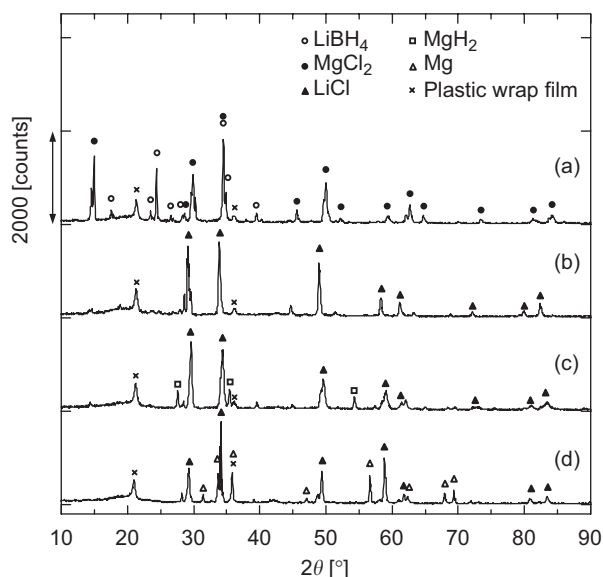
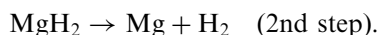
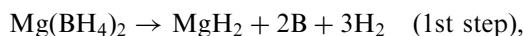


Fig. 5. Powder X-ray diffraction profiles of LiBH₄ and MgCl₂ mixture: (a) before heat treatment; (b) after heat treatment at 593 K < 10 MPa of hydrogen; (c) after desorption up to 593 K; and (d) after desorption up to 773 K in vacuum, respectively.

Fig. 5 shows the XRD measurement results of the LiBH₄ and MgCl₂ mixture (a) before heat treatment, (b) after heat treatment at 593 K for 3 h, (c) after TPD measurement up to 593 K and (d) after TPD measurement up to 773 K. As it is shown in Fig. 5(c), after the desorption measurement up to 593 K, the peaks which correspond to MgH₂ are observed, whereas the diffraction pattern of metallic Mg is not observed. On the other hand, as is shown in Fig. 5(d), after heated up to 773 K, the peaks corresponding to Mg are clearly present, where the peaks from MgH₂ are no longer observed. This result supports the following two-step reaction:



4. Conclusions

We intended to synthesize Mg(BH₄)₂ by decomposition reaction of LiBH₄ with MgCl₂ by heat treatment without

using a solvent, where the product consists of LiCl and a compound of magnesium, boron and hydrogen. Hydrogen desorption property of the product was examined by TPD measurement. It was found that the decomposition temperature of the product is approximately 100 K lower than that of LiBH₄ and the decomposition consists of a two-step reaction. The 1st decomposition peak is dominant and is around 563 K. The 2nd decomposition reaction occurs at the temperature > 590 K. The products of the 1st and 2nd decomposition reaction are MgH₂ and Mg, respectively. Therefore, this is an evidence for Mg(BH₄)₂ as the possible reaction product, although no diffraction pattern was found.

Acknowledgments

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References

- [1] Züttel A. *Materialstoday* 2003;24.
- [2] Bogdanovic BM, Schwickardi J. *J Alloys Compds* 1997;253–254:1.
- [3] Züttel A, Rentsch S, Fischer P, Wenger P, Sudan P, Mauron Ph, et al. *J Alloys Compds* 2003;356:515.
- [4] Chen P, Xiong Z, Luo J, Lin J, Tan KL. *Nature* 2002;420:302.
- [5] Vajo JJ, Skeith SL, Mertens F. *J Phys Chem B* 2005;109(9):3719.
- [6] Orimo S, Nakamori Y, Züttel A. *Mater Sci Eng B* 2004;108:51.
- [7] Nakamori Y, Orimo S. *J Alloys Compds* 2004;370:271.
- [8] Nakamori Y, Miwa K, Ninomiya A, Li H, Ohba N, Towata S, Züttel A, Orimo S. *Communicated*.
- [9] Miwa K, Ohba N, Towata S, Nakamori Y, Orimo S. *J Alloys Compds* 2005;404–406:140.
- [10] Yoshino M, Komiya K, Takahashi Y, Shinzato Y, Yukawa H, Morinaga M. *J Alloys Compds* 2005;404–406:185.
- [11] Plešek J, Hermanek S. *Collect Czech Chem Commun* 1966;31:3845.
- [12] Konoplev VN. *Russ J Inorg Chem* 1980;25(7):964.
- [13] Konoplev VN, Silina TA. *Russ J Inorg Chem* 1974;19(9):1383.
- [14] Stasinevich DS, Egorenko GA. *Russ J Inorg Chem* 1968;13(3):341.
- [15] Orimo S, Nakamori Y, Kitahara G, Miwa K, Ohba N, Towata S, et al. *J Alloys Compds* 2005;404–406:427–30.